

# Analysis of Rigid-Body Effects on Bi-Element Targets

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# Analysis of Rigid-Body Effects on Bi-Element Targets

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#### **Abstract**

The use of semi-infinite bi-element targets in depth of penetration (DOP) tests initially arose from the need to rank performance of ceramic materials under ballistic impact. However, since ceramics exhibit complex damage responses, interpretation of DOP results for ceramic/metal target combinations can be difficult and sometimes misleading. Thus, recent work utilized bi-element metal/metal targets to determine additional mechanisms present in the earlier DOP ceramic/metal target responses. This demonstrated that significant target interactions are present in either combination in addition to specific damage mechanisms inherent in the ceramic response. The target interactions considered before included shock-induced transient effects at the front target surface and shock wave reflections at the target/target interface. In current work, which considers low-density/low-strength target materials, it has been found that rigid-body penetration is present and needs to be taken into account also. This report investigates rigid-body penetration. The work explores the previously cited mechanisms through experimental work and includes a model to explain results.

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### 1. Introduction

The use of semi-infinite bi-element targets of ceramic/metal in depth of penetration (DOP) testing arose from the need to rank ceramic materials. Performance is measured by the DOP of a long-rod penetrator into a semi-infinite steel back plate after passing through a ceramic applique. The penetrator velocity is held constant while the areal density/thickness of the ceramic is varied over a wide range of values. The DOP vs. applique thickness experiments generate performance maps that provide a means to compare performance of various ceramics in armor designs.

The DOP test method has gained acceptance as a valuable tool for comparative testing and ranking of ceramics. However, previous experimental and analytical work has indicated that such results can be difficult to interpret and sometimes misleading. For example, prior work (Rupert and Grace 1993; Grace and Rupert 1993) has identified a dynamic target interaction effect that can alter perceived performance in a manner similar to the known damage mechanisms that occur in ceramics subject to ballistic impact. The interactions include the shock transient associated with penetrator impact on the target front surface and shock wave reflections at target interfaces. In the previous analysis that considers titanium (Ti), as a surrogate for ceramics, and rolled homogeneous armor (RHA) steel, the shock effects were referred to as a "density effect mechanism" for both metallic and ceramic appliques. The current work considers low-density/lower strength target materials of Ti alloy, as a surrogate for ceramic appliques, and aluminum (Al) second element. The experiments show that rigid-body penetration was present and needed to be taken into account. As with the density effect, the appearance of rigid-body penetration can alter perceived target performance substantially. This report investigates the effects of rigid-body penetration.

### 2. Materials

**2.1 Ti.** Since the introduction of Ti and Ti alloys in the early 1950s, these materials have in a relatively short time become the backbone materials for the aerospace, energy, and chemical

industries (Bomberger, Froes, and Morton 1985). The combination of a high strength-to-weight ratio, excellent mechanical properties (i.e., strength vs. temperature), and corrosion resistance makes Ti the best material for many critical applications. However, the traditional high cost of Ti alloys has limited their use to applications for which lower cost materials, such as Al and steel, could not be used.

Ti-6A1-4V alloy dominates structural casting applications. This alloy similarly has dominated wrought industry products since its introduction in the early 1950s, becoming the benchmark alloy against which others are compared (Eylon, Nekmlan, and Thorne 1990). With the recent reduction in the cost of Ti alloys, a renewed interest in using Ti as an armor material is taking place. Property data measured from armor plates used in the recent evaluation of low-cost Ti-6A1-4V plates are listed in Table 1.

**Table 1. Computational Material Properties** 

	DU Alloy	Ti 6A1-4V	7039 A1
Density (ρ) Nominal Strength (S) Yield Strength (Y <sub>ys</sub> ) Young's Modulus (E) Plastic Modulus (E <sub>t</sub> ) Sound Velocity	18.6 g/cm³ 1.38 GPa NU NU NU NU NU	4.45 g/cm <sup>3</sup> 0.91 GPa 0.86 GPa 113.8 GPa 1.9 GPa 6,070 m/s	2.73 g/cm <sup>3</sup> 0.45 GPa 0.48 GPa <sup>a</sup> 75 GPa <sup>a</sup> 0.55 GPa <sup>a</sup> 5,240 m/s

<sup>&</sup>lt;sup>a</sup> Zook, Frank, and Silsby 1992.

NU - Not Used

2.2 Al. Interest in Al alloy armor evolved early in World War II from the testing of 2024-T6 and 7075-T6 Al plates (Mascianica 1979). The two alloys showed good fragmentation protection against high-explosive shells and, in some cases, against armor-piercing ammunition. Later, the Al industry cooperated with the Army in developing 7039 Al to military specification MIL-A-46063 (Materials Directorate 1992). This alloy demonstrated improved protection against kinetic energy ammunition. Since then, 7039 Al has become the standard Al armor for the Army. Property data measured from random plates used at the U.S. Army Research

Laboratory (ARL) (formerly the Ballistic Research Laboratory [BRL]) over the past 10 yr are also listed in Table 1.

## 3. DOP Testing

DOP testing was developed as a means of ranking ceramic materials for ballistic applications (Woolsey, Mariano, and Kokidko 1989; Alme and Bless 1989a, 1989b; Bless, Rosenberger, and Yoon 1987; Woolsey, Mariano, and Kokidko 1990; Frank 1981). Performance is measured by the DOP of a long-rod penetrator into a semi-infinite steel back plate after passing through a ceramic applique. Ceramic performance comparisons are then made between selected baseline materials. We have extended this type of testing to include bi-element metallic targets.

- **3.1 Projectiles.** The projectile used in this study was the 65-g, U-0.75% Ti, long-rod penetrator manufactured by Nuclear Metals, Incorporated. The penetrator had a diameter of 7.70 mm and a length-to-diameter (L/D) ratio of 10. Nominal material properties for these penetrators are as follows: density 18.6 g/cm<sup>3</sup>, hardness R<sub>c</sub> 38-44, yield strength 800 MPa, ultimate strength 1,380 MPa, and elongation 12% (Leonard, Magness, and Kapoor 1992).
- 3.2 Range Setup. The penetrators were fired from a laboratory gun consisting of a Bofors 40-mm gun breech assembly with a custom-made 40-mm smoothbore barrel. The gun was positioned approximately 3 m in front of the targets. High-speed (flash) radiography was used to record and measure projectile pitch and velocity. Two pairs of orthogonal x-ray tubes were positioned in the vertical and horizontal planes along the shot line (as illustrated in Figure 1). Propellant weight was adjusted for desired nominal velocity of 1,500 m/s. Projectiles with a striking total yaw in excess of 2° were considered a "no test," and those data were disregarded.
- 3.3 Target Construction. Targets were multihit targets nominally 152.2 mm  $\times$  304.4 mm (6 in  $\times$  12 in) in size. The first element consisted of a single plate of Ti mechanically clamped to

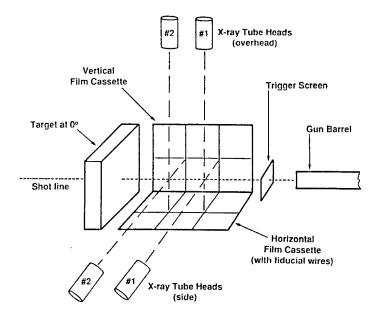


Figure 1. Test Setup.

the second element. Al second elements were constructed from a stack of 76.2-mm (3 in)-thick 7039, MIL-A-46063 Al plates (two to four plates).

## 4. Test Results

**4.1 Baseline Monolithic Ti Data.** Monolithic Ti-6A1-4V penetration data for the depleted uranium (DU) penetrator against monolithic Ti-6A1-4V are based on nine tests, where impact velocities ranged from 1,100 m/s to 1,950 m/s (Burkins 1996). Over this range, the data are linear. Thus, an empirical fit to the data was derived in the following form:

DOP<sub>(Ti)</sub> = 0.0949 
$$\left(\frac{\text{mm·s}}{\text{m}}\right) \cdot V_s - 56.7 \text{ mm},$$
 (1)

where  $V_s$  is the striking velocity in meters/second, and semi-infinite DOP and the constant are in millimeters. Residual penetrator lengths were not measured during these tests.

**4.2** Baseline Monolithic Al Data. Monolithic 7039 Al data for the DU penetrator against Monolithic 7039 Al are limited to 17 tests listed in Appendix A, with impact velocities ranging from 500 m/s to 2,000 m/s (Rupert 1994). Between 1,000 m/s and 2,000 m/s, the data are linear. An empirical fit to the 11 data points was derived as shown:

$$DOP_{(Al)} = 0.1195 \left(\frac{mm \cdot s}{m}\right) V_s - 6.89 \text{ mm},$$
 (2)

where  $V_s$  is the striking velocity in meters/second, and the semi-infinite DOP and constant are in millimeters. In order to correct for variations in the actual striking velocities, all residual penetration values for metallic bi-element targets were adjusted to a striking velocity of 1,500 m/s by the following correction based on equation (2):

$$DOP_{(Al)} = Measured DOP_{(Al)} + 0.1195 \left(\frac{mm \cdot s}{m}\right) (1,500 \text{ m/s} - V_s). \tag{3}$$

These corrections were made as to minimize the scatter in the bi-element DOP data resulting from round-to-round velocity variations.

4.3 Ti/Al Data. Corrected DOP results for Ti/Al bi-element targets are shown in Figure 2 and listed in Appendix B. (A second-order linear regression curve was fitted to the data using Sigma Plot 5.0 automatic plotting function.) Examination of the regression curve for the ballistic data and the rule of mixtures shows a similar trend as found in the Ti/RHA data (Rupert and Grace 1993).

The rule of mixtures for the study takes the following mathematical form:

DOP = DOP<sub>(2)</sub> 
$$\cdot \left[1 - \frac{T_{(a)}}{DOP_{(1)}}\right],$$
 (4)

where  $DOP_{(2)}$  is the semi-infinite DOP value for the second element,  $DOP_{(1)}$  is the semi-infinite DOP value for the first element, and  $T_{(a)}$  is the applique thickness. Implicit in the rule of mixtures are the following assumptions:

- (1) The performance as measured by depth of penetration of the two target elements is linearly additive; there are no interactions or synergistic effects associated with the bi-element target.
- (2) The ballistic efficiency of the rear element is constant and independent of the intermediate penetrator length and velocity at the interface between the two elements.
- (3) Velocity corrections for the bi-element target are equivalent to velocity corrections for a semi-infinite target of the rear element.

The density effect does not account for the shift of the data up and to the right as in the previous case. However, unlike the RHA/Ti bi-element targets, there are substantial differences in strength, density, and sound velocity of the two metals. Primarily as a result of the aluminum's lower strength and lower density when compared to Ti and RHA, rigid-body penetration within the rear element was introduced to the problem.

## 5. Modeling

To investigate the performance of Ti/Al targets, two different models were used. One treats target penetration during an initial phase where the rod undergoes erosion, while the other treats target penetration during a subsequent phase where the rod remains rigid. The terms used to differentiate the two processes are "eroding-body" penetration and "rigid-body" penetration, respectively. The quoted qualifiers refer to the state of the penetrator during target penetration.

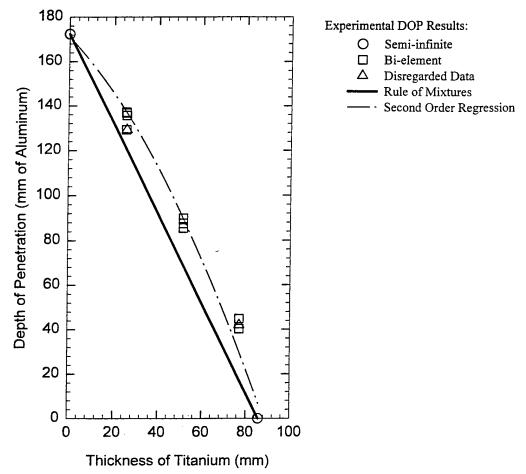


Figure 2. Ti/Al Ballistic Results.

To analyze the dynamic effects that develop and the eroding-body penetration, the nonsteady penetration development of Grace (1993) and its application to bi-element targets by Grace and Rupert (1993) were utilized. There, long-rod penetrators impacting semi-infinite and bi-element targets were considered. For the bi-element targets of interest here, the overall target is semi-infinite and layered as well. The geometry of the bi-element target is shown in Figure 3. Impact conditions are rod impact velocity  $v_s$ , initial rod length  $\ell_o$ , and first-element thickness  $a_o$ . The backup target, or second element, is semi-infinite metal. Impact conditions for the second layer depend upon rod quantities that exist after penetration through the first layer. These are defined as rod velocity  $v_1$  and rod length  $\ell_1$ . As an extension to previous work (Grace and Rupert 1993), it is assumed that the total penetration  $P_T$  in the overall target is the sum of that through each element. This gives

$$P_{T} = - \int_{\ell_{0}}^{\ell_{1}} \left( \frac{u}{v - u} \right)_{1} dl - \int_{\ell_{1}}^{\ell_{2}} \left( \frac{u}{v - u} \right)_{2} dl + P_{RB}, \qquad (5)$$

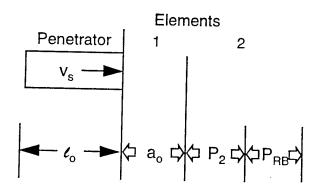


Figure 3. Bi-Element Target Geometry.

where  $[u/(v-u)]_1$  and  $[u/(v-u)]_2$  are respective penetration velocities divided by the respective penetrator material flow rates for the two elements within the eroding-body phase, and  $P_{RB}$  is the subsequent rigid-body penetration within the second element. When the penetrator can overmatch the first element, the first integral on the right-hand side of equation (5) is equal to  $a_0$ . However, it is necessary to calculate the penetration through the first element to arrive at penetrator length  $\ell_1$  and erosion rates  $[u/(v-u)]_2$  to be used in the second integral. Also, when rigid-body penetration is involved, the limit  $\ell_2$  on the second integral will be determined when rod erosion stops. This will result from the condition  $(v-u)_2 = 0$ , which is given by previous work (Grace 1993), or by a rod-erosion cutoff velocity to be defined subsequently. The DOP or residual penetration  $P_r$  into the backup element is given by the second integral plus penetration due to the rigid-body contribution, or

$$P_{r} = -\int_{\ell_{1}}^{\ell_{2}} \left(\frac{u}{v-u}\right)_{2} dl + P_{RB}.$$
 (6)

Rod erosion v - u and target erosion u (penetration rate) were given respectively as

$$v - u = (v_s - u_o) \left[ 1 + \frac{2S_p}{\rho_p (v_s - u_o)^2} \ln \left( \frac{\ell}{\ell_o} \right) \right]^{1/2}$$
 (7)

and

$$u = u_o \left[ 1 + \frac{2S_t}{\rho_t u_o^2} ln \left( \frac{\ell}{\ell_o} \right) \right]^{1/2}$$
 (8)

When solving equation (5), penetration into the first element as given by the first integral is calculated stepwise as if the element were semi-infinite. The process continues up to the point where the penetration depth reaches  $a_0$ . This provides the starting conditions  $v_1$ ,  $u_1$ , and  $\ell_1$ , for the second integral. Since equations (7) and (8), as written, apply to the first element, their use for the second element requires  $v_s$ ,  $u_0$ , and  $\ell_0$  to be replaced with  $v_1$ ,  $u_1$ , and  $\ell_1$ . The second integral is calculated stepwise up to the point where the rod stops eroding, which provides penetration depth  $P_2$  into the second element during the eroding-body phase, with rod length  $\ell_2$  and its velocity  $v_2$  as starting conditions for the rigid-body penetration portion of the problem.

Penetration into the first element was calculated using previous methods (Grace and Rupert 1993) that account for the shock transient due to impact at the target front surface and shock wave reflections, due to density and sound velocity changes across the target material interface. Treating the first element as semi-infinite produces a penetration process that ignores possible influences, due to the properties of the backup material. A model was developed to account for the density change across the target material interface and to explore its ability to match the experimental observations (Rupert and Grace 1993). This model uses a simplified version of one-dimensional shock wave propagation to treat the influence on penetration due to shock reflection from a proximate interface. Figure 3 depicts the penetrator/target and bi-element target interfaces of interest. An upper limit for the penetration rate is taken to be the particle velocity u<sub>s</sub> associated with the shock wave that is generated by penetrator impact with element 1. Two well-known relations from the theory of shock wave propagation give the pressure p, shock velocity U<sub>s</sub> and particle velocity u immediately behind the shock and density ρ as where c is a

$$p = \rho u U, \qquad U = c + gu, \qquad (9)$$

velocity of sound, and g is a material constant. Applying these two equations to the penetrator/target and bi-element target interfaces together with appropriate boundary conditions gives the following expressions used in the current model as

$$u_{s} = \frac{\rho_{p}/\rho_{l}}{1 + \rho_{p}/\rho_{l}} v_{s}, \qquad u_{r} = (\rho_{l}/\rho_{s}) u_{i},$$
 (10)

under simplifications that the sound speeds of the penetrator and targets are taken to be equal, and the variation of shock speed with particle velocity has been ignored. In equation (9),  $\rho_p$  is the rod density,  $u_r$  is the velocity of material reflected from the interface, and  $u_i$  is the incident material velocity. Upon impact, the initial penetration rate at the front surface  $u_s$  drops to a quasi-steady value  $u_o$  as penetration proceeds to a depth on the order of a penetrator diameter. The model permits the penetration rate to be increased or reduced from  $u_o$ . The change has the following form:

$$u_e = u_o + q(u_s - u_o),$$
 (11)

where  $u_e$  is the effective penetration rate,  $q(u_s - u_o)$  represents an increment of velocity change, and  $u_o$  is the rate given by previous theory (Grace 1993, equation [25]). The form of q is arbitrary and is chosen for convenience to include influences generated by the transient and bielement target interface as

$$q = \pm k \left(\frac{\rho_2}{\rho_1}\right) \left(\frac{DOP_{(1)} + d - a_0}{DOP_{(1)}}\right)^2, \qquad (12)$$

where d is rod diameter, and  $DOP_{(1)}$  is the semi-infinite DOP value for the first-element material. The last term on the right-hand side of equation (12) allows the correction to decrease as the reflective wave weakens due to increased distance to the reflective boundary. The ratio of densities appearing in equation (12) takes into account the strength of the reflected wave as

indicated by equation (10), and the sign change indicates the direction of material flow. The value for k is chosen so that q does not exceed q = 1 and the penetration rate of equation (11) does not exceed  $u_s$ . Equations (5), (7), (8), and (11) give the penetration through the first element and the expected rod length and velocity to be used as starting values in the calculation for DOP as given by equation (6). The final penetrator length  $\ell_2$  required in the integration of equation (6) is given by the nonsteady penetration theory (Grace 1993).

As indicated, the nonsteady penetration development (Grace 1993) provides the needed parameters at the point where rigid-body penetration begins, but does not account for rigid-body penetration itself. Thus, beyond the penetration contribution given by the second integral,  $P_2$ , it is necessary to calculate  $P_{RB}$  separately. For present purposes,  $P_{RB}$  is determined using the Alekseevshii/Tate penetration algorithm for rigid-body penetration, as presented by Zook, Frank, and Silsby (1992), and takes the following form:

$$P_{RB} = \frac{\rho_{p} \ell_{2}}{2k_{t} \rho_{t}} \ln \left[ 1 + \frac{k_{t} \rho_{t} v_{2}^{2}}{H} \right]. \tag{13}$$

The Alekseevshii formulation treats k<sub>t</sub> as a shape factor. Tate takes the value for k<sub>t</sub> to be 0.5, corresponding to the value that appears in the Bernoulli equation. The target resistance pressure H of 2.21 GPa was calculated from Goodier's expanding spherical cavity analysis (Goodier 1965). Accordingly, the target resistance is as follows:

$$H = \frac{2Y_{ys}}{3} \left[ 1 + \ln \left( \frac{2E}{3Y_{ys}} \right) \right] + \frac{2\pi^2}{27} E_t, \qquad (14)$$

where  $Y_{ys}$  is the target yield strength, E is Young's modulus, and  $E_t$  is the slope from the yield point to the ultimate strength point, assuming a bilinear stress-strain behavior curve. Property values used in the modeling are given in Table 1.

In the present calculations, rod erosion was assumed to stop when penetration velocity reached a critical value. That value, u<sub>c</sub>, is determined when the pressure on the nose of the penetrator drops below the stress required for rod erosion. The total stress on the rod, due to its flow into the target, is the sum of the dynamic pressure due to its velocity plus the strength of the target, or

$$p_{c} = \frac{1}{2} \rho_{t} u_{c}^{2} + S_{t} . \qquad (15)$$

Taking values  $S_t = 0.625$  GPa for the model's target material dynamic strength and  $p_c = 1.38$  GPa for the model's penetrator dynamic strength gives an erosion cutoff velocity of 743 m/s for DU rods into Al targets. Thus, the second integral was solved stepwise until the penetration velocity was reduced to 743 m/s. This gave rod length  $\ell_2$ , velocity  $v_2$ , and penetration  $P_{RB}$  in the Al at the end of the eroding-body penetration phase.

## 6. Experimental and Model Results

The current work addresses low-density/lower strength target materials of Ti/Al and the possibility of rigid-body penetration in these targets. The experimental results provided three observations suggesting that at least some rigid-body penetration was taking place in the Al backup targets:

- (1) The greater DOPs than expected (based on erosion) in the Al suggested that the penetrator had higher efficiency than it would have had otherwise. This is consistent with rigid-body penetration.
- (2) Within the DOP tests, recovered residual penetrators had lengths of 2.0–2.5 rod diameters, whereas 1.0–1.5 factors would be expected under eroding-body penetration only.

(3) Most convincingly, experimental data for penetration of DU rods into semi-infinite aluminum with impact velocities between 500 m/s to 750 m/s exhibited rigid-body penetration, exclusively. Complete uneroded penetrators were recovered from the Al targets.

Figure 4 provides DOP data as a function of Ti applique thickness. The data points on the abscissa and ordinate correspond to semi-infinite targets of penetration by the DU penetrator into Ti (86.4 mm) and Al (172.4 mm), respectively. Penetration calculations were carried out for each of the two cited semi-infinite targets. Results indicated that the semi-infinite Ti was penetrated by eroding-body penetration only because of its higher strength and density as compared to Al. Using nominal strengths for DU and Ti from Table 1 gave a penetration depth into semi-infinite Ti of 85.7 mm. On the other hand, penetration into the Al target resulted from both eroding-body and rigid-body penetration phases. For the Al, the nonsteady theory gave an eroding-body penetration of 138 mm and a rod length of 16.7 mm at the erosion cutoff velocity of 743 m/s. These values and equation (13) provided a calculated rigid-body contribution of 36 mm for a total penetration into Al of 174 mm.

For the bi-element targets, calculations indicated that both eroding-body and rigid-body penetrations were present in the backup Al. Again, 743-m/s erosion cutoff velocity was used. Calculated values for residual penetrator length (rigid-body length) over the range of Ti thicknesses were from 18.7 to 15.2 mm, while the experimentally measured average values varied between 19 and 10 mm. The DOP calculations are presented in Figure 4. The straight solid line connecting the semi-infinite points (Figure 4, curve 3) represent expected results from the rule of mixtures equation (4). The first point to note is that the calculated eroding-body phase (Figure 4, curves 4 and 5) gave DOPs that approach the rule-of-mixture curve for the thicker Ti appliques. The eroding-body calculations are much further beneath the rule of mixture when greater amounts of Al are penetrated in the bi-element target with the thinner Ti sections. The amount of rigid-body penetration is constant and not proportional to the Ti applique thickness as a result of equation (15). While this proposition would presently be difficult to confirm experimentally, the constant cutoff velocity assumed combined with the nearly constant residual

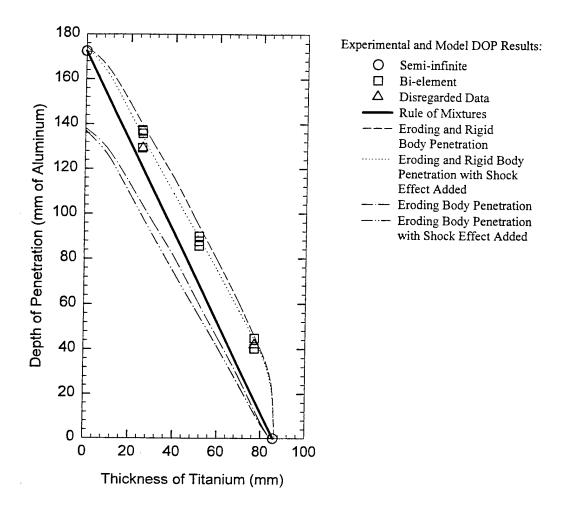


Figure 4. Ti/Al Data and Model Results.

penetrator lengths measured support this proposition as a strong possibility. Additional experimental refinements are being investigated to determine the transition between penetration modes. As expected, there was a slight downward shift of about 6–9% when the shock wave or density effect was taken into account (Rupert and Grace 1993; Grace and Rupert 1993). The upper two curves (Figure 4, curves 1 and 2) represent the total calculated penetration based on eroding-body and rigid-body contributions. Shock effects were not accounted for in the higher curve (Figure 4, curve 1), but were taken into account in the lower one (Figure 4, curve 2). In either case, the rigid-body penetration was about 30% of the total in the backup for thin Ti appliques. It became the major contribution for thick Ti appliques. For the low-strength Al targets considered here, it was necessary to account for rigid-body penetration.

## 7. Summary and Conclusions

The experiments and penetration analysis provided relative contributions of eroding-body and rigid-body penetration phases. Further, the analysis showed that calculated shock effects in these particular Ti/Al targets influence penetration by about 6–9%. The amount of rigid-body penetration in the backup Al target appears to be nearly constant throughout the range of Ti applique thickness. This accounted for about 30% for thin sections and most of the penetration where the applique was thick. The remainder or initial penetration into the rear element was generated during the rod erosion-based phase. Eroding-body penetration in the backup Al was highest for thin appliques, but contributed little for the thickest ones when the residual rod velocity at the interface approached the cutoff velocity for the aluminum. It is believed that targets having low strength relative to the penetrator will respond in the same fashion, generally. Therefore, the present findings should be applicable to a number of lightweight armor systems, to include composite armor designs.

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Appendix A:

Al Data

Table A-1. Penetration Results for 7039 Al

Striking Velocity (m/s)	Pitch (deg)	Yaw (deg)	DOP (mm)
576	1.00U	0.25L	60.9
741	1.00U	0.25L	125.9
807	1.50U	0.50R	114.4
911	0.50D	0.25R	96.0
1,000	0.50D	0	99.8
1,098	<b>0.25</b> U	0	114.1
1,146	0.50D	0.50L	125.1
1,184	0	0.50L	130.8
1,296	1.25D	1.00R	147.5
1,502	0	0.50R	174.1
1,505	1.00U	0.75R	176.1
1,511	0	0.50R	174.8
1,513	1.00D	1.25R	174.8
1,515	0	1.50R	176.7
1,718	1. <b>75</b> D	0.25L	197.8
2,000ª			223.1
2,013	0.25U	0.75L	230.1

<sup>&</sup>lt;sup>a</sup> Estimated velocity from powder curve.

Appendix B:

Ti/Al Data

Table B-1. DOP Results for Ti/Al

Applique Thickness (mm)	Striking Velocity (m/s)	Pitch (deg)	Yaw (deg)	DOP (mm)	Corrected DOP (mm)
25.5	1,354	1.00D	0.75R	111.9	129.3
25.5	1,514	1.50D	0	138.8	137.2
25.7	1,516	1.50D	0.50L	137.6	135.7
25.7	1,522	0.25D	2.25R	139.3	136.7
25.5	1,632	<b>0.50</b> U	0.75R	152.6	136.8
51.5	1,508	0.75D	1.00R	86.5	85.6
51.5	1,511	0.75D	0.75R	89.0	87.7
51.5	1,499	1.00U	0	89.9	89.9
77.0	1,524	0	1.00L	43.2	40.3
77.0	1,524	0	2.50R	44.9	42.0
77.0	1,518	1.75D	1.25R	47.0	44.8

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The use of semi-infinite I performance of ceramic mate interpretation of DOP results recent work utilized bi-elem ceramic/metal target response in addition to specific damag included shock-induced transinterface. In current work, we penetration is present and nework explores the previously	erials und s for ceran ent meta es. This c ge mechan scient effe which conseeds to be	er ballistic impact. Ho mic/metal target combi al/metal targets to deter demonstrated that significant in the celects at the front target siders low-density/low-e taken into account al	wever, since ceramics nations can be difficult rmine additional mechificant target interaction ramic response. The test surface and shock we strength target materialso. This report investigation	exhibit t and so anisms are p arget invave refus, it has igates r	present in the earlier DOP resent in either combination teractions considered before elections at the target/target is been found that rigid-body igid-body penetration. The
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